

Magnetotransport And Spin Dynamics Investigations On The Te-based Topological materials And Their Heterostructures Fabricated By Sputtering

Lalit Pandey



DEPARTMENT OF PHYSICS

INDIAN INSTITUTE OF TECHNOLOGY DELHI

JUNE 2023

**Magnetotransport and Spin Dynamics Investigations on
the Te-based Topological materials and their
Heterostructures Fabricated by Sputtering**

by

Lalit Pandey

DEPARTMENT OF PHYSICS

Submitted

in the fulfilment of the requirement of the degree of the Doctor of Philosophy

to the



INDIAN INSTITUTE OF TECHNOLOGY DELHI

JUNE 2023

© **Indian Institute of Technology Delhi (IITD), New Delhi, 2023**

Dedicated
to
My Parents
and
My lovely Sisters

Certificate

This is to certify that the thesis entitled “**Magnetotransport and Spin Dynamics Investigations on the Te-based Topological materials and their Heterostructures Fabricated by Sputtering**”, which is being submitted by **Mr. Lalit Pandey** to the **Indian Institute of Technology Delhi**, New Delhi, for the award of the degree of **Doctor of Philosophy** in Physics, is a record of bonafide research work carried out by him. He has worked under my supervision and guidance and has fulfilled the requirements for the submission of this thesis, which, in my opinion, has reached the requisite standard.

The results contained in this thesis have not been submitted, in part or full, to any other University or Institute for the award of any degree/diploma.

Prof. Sujeet Chaudhary

Professor

Department of Physics

Indian Institute of Technology Delhi

New Delhi-110016

India

Date:

Acknowledgments

I wish to extend my utmost gratitude to my Ph.D. supervisor, Prof. Sujeet Chaudhary, who provided me with the opportunity to work in the Thin Film Laboratory (TFL) under his guidance. I am sincerely grateful to him for allowing me the freedom to test out any ideas and giving me permission to fail while experimenting. Throughout my Ph.D. journey, he posed challenging questions on research topics that helped me to comprehend complex and abstract concepts related to the research problem. I would also like to thank him for the training he provided me during the initial stages of my Ph.D., which helped me acquire essential experimental skills. I am thankful for his constant encouragement, constructive criticism, and unwavering efforts, which significantly improved my abilities as a researcher.

I would also like to acknowledge the Department of Physics at IIT Delhi for offering me the opportunity to pursue my Ph.D. and the University Grants Commission, Government of India, for providing me with the research fellowship throughout my Ph.D. tenure.

I am grateful to my SRC committee members, Prof. Ratnamala Chatterjee, Prof. Rajendra Singh Dhaka, and Prof. Jayant Jain, for their periodic feedback, insightful comments, and fruitful discussions during the comprehensive seminar, various progress meetings, and pre-Ph.D. seminar.

I would like to express my gratitude to Prof. Peter Svedlindh, Prof. Biplap Sanyal and Dr. Xin Chen for their collaboration and contributions to our work. I am also thankful to my seniors - Dr. Sajid Husain, Dr. Vineet Barwal, Dr. Soumyarup Hait, Dr. Nanhe Kumar Gupta, Dr. Rahul Gupta, and Dr. L. Saravanan, for guiding me like an elder brother throughout my Ph.D. journey.

Special thanks go to Prof. Pankaj Srivastava, Prof. Pranaba Kishor Muduli, and Prof. Ratnamala Chatterjee for giving me the opportunity to independently handle the departmental and central research facilities, such as X-ray Diffractometer and Physical Property Measurement System. Without these facilities, the experimental work performed for this thesis would not have been possible. I also acknowledge the Nanoscale Research Facility (NRF) for providing the excess to the excellent experimental facilities.

I would like to express my heartfelt appreciation to Shri Nagendra Chaudhary, Dr. Dinesh Dixit, and Dr. Veer Singh for providing invaluable technical support throughout my research work.

My labmates and friends at TFL - Vireshwar Mishra, Nakul Kumar, Nikita Sharma, Naina Kushwaha, Sanjay Kumar Kedia, Amar Kumar, Kaushalya, Akashdeep, Kriti Jain, Deepak, Badsha, Amir, Harjinder, Ravi Meena, Ravi Raturi, Sarita, Avantika, Sarah, Ekta Goyat,

Pradeep Sharma, Shubhashish, Abhay - provided me with a wonderful atmosphere to pursue my research work and engage in scientific discussions, which have been instrumental in enhancing my knowledge and skills immensely.

I would also like to thank my friend Vipul and my flatmates Vishal, Ravinder, and Sharad for making this Ph.D. journey memorable.

Lastly, I would like to express my sincere gratitude to my family members, my father Mr. Anil Kumar Pandey, my mother Mrs. Manorama Pandey, and my sisters Anupama and Nirupama Pandey. I will always remain thankful for their unconditional love, support, and encouragement. Their dedication and sacrifices have been instrumental in shaping my values and beliefs, and I am grateful for their constant support in all aspects of my life.

Lalit Pandey

(June 2023)

Abstract

Topological materials (TMs) belong to the class of quantum materials whose physical properties, unlike the usual classical materials, cannot be explained by the theory of spontaneous symmetry breaking. They are classified based on a topological invariant (ν), which is defined by the band structure topology. A material is considered topological if it has a non-zero value of ν , which can be induced by introducing high spin-orbit coupling (SOC) to a previously non-topological material with $\nu = 0$. This can cause an inversion of conduction and valence bands, leading to a quantum phase transition. During this transition, topological surface states (TSSs) emerge, which exhibit unique properties such as Dirac Fermionic nature, helical spin polarization, and forbidden backscattering of electrons. These states are symmetry protected, making them robust against scattering off from the non-magnetic impurities. Literature reviews on spin-orbit torque devices based on the heterostructure of TMs and ferromagnets (FMs) suggest that TMs possess high values of spin-charge conversion efficiency, which can exceed 50%, resulting in much lower values of switching current compared to those in the heavy metals. Therefore, TMs have attracted attention for their exotic properties and high spin-charge conversion efficiencies, making them promising materials for microelectronics and spintronics applications. Heterostructures composed of TMs and other materials, such as FMs and superconductors, exhibit interesting phenomena such as the quantum anomalous Hall effect, topological superconductivity, and topological magnetoelectric effect. These structures can also host quasiparticles associated with Majorana bound states, Weyl Fermions, and magnetic monopoles, among other striking features and phenomena.

Even having these many exotic properties, the TMs face several difficulties, which hinder the integration of TMs in future microelectronic devices. Some of these difficulties are the absence of a growth procedure to synthesize excellent quality and stoichiometric TMs using industrial-friendly sputtering technique and simple investigation protocol to evaluate topological materials using easily accessible magneto-transport measurements. This thesis is aimed to possibly explore the solutions to these difficulties. For this, the Bi_xTe_y (BT) topological system is grown on Si(100) using co-sputtering. Firstly, the control of altering the Te composition over a wide range in a systematic manner is achieved by closely examining the changes in the structural and compositional analyses vis a vis the growth conditions. Then, by using the same recipe employed in synthesising Bi_xTe_y , thin films of varieties of other TMs and 2D FMs such as Sb_2Te_3 , Cr_2Te_3 , BiSb , NiTe_2 , PtTe_2 , PdTe_2 and PdBi_2 are successfully synthesized using DC-magnetron sputtering technique only.

For the second challenge, highly oriented Bi_2Te_3 thin films are synthesized on the Al_2O_3 (0001) substrate using the same co-sputtering technique. The detailed investigation of the temperature and magnetic field dependent transport properties of these high quality Bi_2Te_3 thin films is undertaken. The different models such as modified Hikami-Larkin-Nagoaka (HLN), Lu-Shen and modified Al'tshuler-Aronov model are used to analyze the temperature dependent, and in-plane and out-of-plane magnetic field dependent transport behaviour. From the analyses of such data, different topological parameters such as coherency factor (α), decoherence factor (p), Berry phase (ϕ_B), mass term (m) are estimated.

After suggesting possible solutions to challenges that hinder the implementation of TMs in future electronic devices, the proposed growth methodology and investigation protocols is utilized in this thesis work to find answer to some fundamental questions. Some of these questions are – what is the influence of topological properties of TMs by tuning the concentration of chalcogenides (Te or Se), and what is the incorporation of magnetic and non-magnetic elements during the growth of topological materials? To answer these queries, first, using the structural properties information on different Te composition Bi_xTe_y binary alloys thin films, the samples are divided into three different Te-concentration phases namely Te-deficient (TD), Te-stoichiometric (TS), and Te-excessive (TE). It is shown using the magneto-transport measurement on these three samples that through the tuning of Te-concentration, one can manipulate the topological properties of TMs. These experimental findings are further supported by first principle theoretical calculations. For the second query, the Bi_2Te_3 films were intercalated with Cr as magnetic element and Au as non-magnetic element. The different topological parameters are estimated by performing and analyzing the magneto-transport data on pristine, Cr-doped and Au-doped Bi_2Te_3 films using the modified HLN and Lu-Shen models. Based on the estimated parameters, it is demonstrated that the topological properties got significantly suppressed in Cr-doped BT films compared to the pristine BT films while slightly enhanced in the Au-doped BT films. This study correlates well to the well-established phenomenon of suppression of topological properties in the presence of magnetic impurities due to the breaking of time reversal symmetry (TRS) in the topological systems.

After answering these fundamental questions, the thesis work is further extended to examine the spin dynamics and spin-charge conversion efficiency in the heterostructure of TMs and FMs fabricated using sputtering techniques. The ferromagnetic resonance (FMR) measurements are performed on these heterostructure samples and the variation of α_{eff} and $\mu_0 M_{eff}$ with the change in thickness of the FM films of $\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ (CFB) in them are investigated. The spin dynamic results show that the effective spin mixing conductance is overestimated in the in-plane FMR due to the two-magnon scattering (TMS) contribution in

damping. From the analysis of the out-of-plane FMR, it is found that the effective spin mixing conductance at room temperature to be 13.13 nm^{-2} and 4 nm^{-2} with and without TMS contribution, respectively. Additionally, the effective spin-mixing conductance value gradually increases with decreasing temperature, which is in accordance with the surface coherence length of BT. These findings show the influence of spin-polarized topological surface states on the magnetization dynamics of ferromagnets when placed adjacent to topological materials. The excellent quality of the sputter-grown ferromagnet-topological insulator heterostructure and its efficient spin-charge conversion performance demonstrate the topological materials' importance in future magnetic memory devices. The spin dynamics measurements are also performed on the $\text{PdTe}_2/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ bilayers. The effective spin mixing conductance and spin current density are found to be $\sim 2 \times 10^{20} \text{ m}^{-2}$ and 2.5 MA m^{-2} , respectively in $\text{PdTe}_2/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ bilayers which are quite larger than those observed in HM/FM bilayers, and comparable to those reported for MBE-grown TI/FM heterostructures.

Thus, this thesis work proposes a growth methodology to synthesize TMs by an industry-friendly sputtering technique which yielded a very good success on making highly oriented and good quality films of varieties of TMs. Moreover, this work provides a unified and comprehensive account of a detailed interpretation of the magnetotransport data of a topological insulator system using different theoretical models available in literature, which is important both from the fundamental and technological development of this field in the future. Further, this thesis demonstrates an effective way to manipulate quantum properties such as topological characteristics of the materials by tuning chalcogenides compositions and by incorporating magnetic and non-magnetic elements during the growth of the topological system. Finally, this thesis demonstrates the efficient spin-charge conversion performance in sputtered grown TMs/FMs bilayers which paves the way to implement topological materials in future spintronics devices.

सार

टोपोलॉजिकल मैटेरियल्स क्वांटम सामग्रियों के वर्ग से संबंधित हैं, जिनके भौतिक गुण, सामान्य मैटेरियल्स सामग्रियों के विपरीत, स्पॉन्तनोउस स्यूम्मेन्ट्रिय ब्रेकिंग के सिद्धांत द्वारा नहीं समझाए जा सकते हैं। उन्हें टोपोलॉजिकल इनवेरिएंट (ν) के आधार पर वर्गीकृत किया गया है, जिसे बैंड संरचना टोपोलॉजी द्वारा परिभाषित किया जाता है। एक सामग्री को टोपोलॉजिकल माना जाता है यदि इसमें ν का गैर-शून्य मान है, जिसे $\nu = 0$ के साथ पहले गैर-टोपोलॉजिकल मैटेरियल्स में उच्च स्पिन-ऑर्बिट कपलिंग पेश करके प्रेरित किया जा सकता है। इससे कंडक्शन और वैलेंस बैंड में व्युत्क्रमण हो सकता है, जिससे क्वांटम फेज ट्रांजिशन हो सकता है। इस ट्रांजिशन के दौरान, टोपोलॉजिकल सरफेस स्टेट्स उभरते हैं, जो डायराक फर्मियोनिक प्रकृति, हेलिकल स्पिन ध्रुवीकरण और इलेक्ट्रॉनों के निषिद्ध बैकस्कैटरिंग जैसे अद्वितीय गुणों का प्रदर्शन करते हैं। ये अवस्थाएं समरूपता से संरक्षित हैं, जो उन्हें गैर-चुंबकीय अशुद्धियों से बिखरने के खिलाफ मजबूत बनाती हैं। टोपोलॉजिकल मैटेरियल्स और फेरोमैग्नेट्स के हेटरोस्ट्रक्चर के आधार पर स्पिन-ऑर्बिट टॉर्क उपकरणों पर साहित्यिक समीक्षा से पता चलता है कि टोपोलॉजिकल मैटेरियल्स में स्पिन-चार्ज रूपांतरण दक्षता के उच्च मूल्य होते हैं, जो 50% से अधिक हो सकते हैं, जिसके परिणामस्वरूप स्विचिंग करंट का मान भारी धातुओं की तुलना में बहुत कम होता है। इसलिए टोपोलॉजिकल मैटेरियल्स ने अपने अद्भुत गुणों और उच्च स्पिन-चार्ज रूपांतरण क्षमता के लिए ध्यान आकर्षित किया है, जिससे वे माइक्रोइलेक्ट्रॉनिक्स और स्पिट्रॉनिक्स अनुप्रयोगों के लिए आशाजनक सामग्री बन गए हैं। टोपोलॉजिकल मैटेरियल्स और अन्य मैटेरियल्स, जैसे फेरोमैग्नेट्स और सुपरकंडक्टर्स से बने हेटरोस्ट्रक्चर, क्वांटम विसंगतिपूर्ण हॉल प्रभाव, टोपोलॉजिकल सुपरकंडक्टिविटी और टोपोलॉजिकल मैग्नेटोइलेक्ट्रिक प्रभाव जैसी दिलचस्प घटनाएं प्रदर्शित करते हैं। ये संरचनाएं अन्य हड़ताली विशेषताओं और घटनाओं के अलावा, मेजराना बाउंड स्टेट, वेइल फर्मियन और चुंबकीय मोनोपोल से जुड़े क्वासिपार्टिकल्स की भी मेजबानी कर सकती हैं।

इतनी सारी अद्भुत गुणों के होते हुए भी, टोपोलॉजिकल मैटेरियल्स को कई कठिनाइयों का सामना करना पड़ता है, जो भविष्य के माइक्रोइलेक्ट्रॉनिक उपकरणों में टोपोलॉजिकल मैटेरियल्स के एकीकरण में बाधा बनती हैं। इनमें से कुछ कठिनाइयाँ औद्योगिक-अनुकूल स्पटरिंग तकनीक और आसानी से सुलभ मैग्नेटो-ट्रांसपोर्ट माप का उपयोग करके टोपोलॉजिकल सामग्रियों का मूल्यांकन करने के लिए सरल जांच प्रोटोकॉल का उपयोग करके उत्कृष्ट गुणवत्ता और स्टोइकोमेट्रिक टोपोलॉजिकल मैटेरियल्स को संश्लेषित करने के लिए विकास प्रक्रिया की अनुपस्थिति हैं। इस थीसिस का उद्देश्य संभवतः इन कठिनाइयों के समाधान का पता लगाना है। इसके लिए, Bi_xTe_y (BT) टोपोलॉजिकल सिस्टम को सह-स्पटरिंग का उपयोग करके Si(100) पर छलरचना किया गया है। सबसे पहले, एक विस्तृत श्रृंखला में टेल्यूरियम संरचना को व्यवस्थित तरीके से बदलने का नियंत्रण विकास की स्थितियों की तुलना में संरचनात्मक और संरचनागत विश्लेषणों में परिवर्तनों की बारीकी से जांच करके प्राप्त किया गया है। फिर, Bi_xTe_y

को संश्लेषित करने में नियोजित उसी नुस्खा का उपयोग करके, अन्य टोपोलॉजिकल मैटेरियल्स और 2D फेरोमैग्नेट्स जैसे Sb_2Te_3 , Cr_2Te_3 , $BiSb$, $NiTe_2$, $PtTe_2$, $PdTe_2$ और $PdBi_2$ की किस्मों की पतली फिल्मों को केवल DC-मैग्नेट्रॉन स्पटरिंग तकनीक का उपयोग करके सफलतापूर्वक संश्लेषित किया गया है।

दूसरी चुनौती के लिए, अत्यधिक उन्मुख Bi_2Te_3 पतली फिल्मों को उसी सह-स्पटरिंग तकनीक का उपयोग करके Al_2O_3 (0001) सबस्ट्रेट पर संश्लेषित किया गया है। इन उच्च गुणवत्ता वाली Bi_2Te_3 पतली फिल्मों के तापमान और चुंबकीय क्षेत्र पर निर्भर ट्रांसपोर्ट गुणों की विस्तृत जांच की गई है। संशोधित हिकामी-लार्किन-नागोका (एचएलएन), लू-शेन और संशोधित अल्टशुलर-अरोनोव मॉडल जैसे विभिन्न मॉडलों का उपयोग तापमान पर निर्भर, और सतह के समानांतर और सतह के लंबवत चुंबकीय क्षेत्र पर निर्भर ट्रांसपोर्ट व्यवहार का विश्लेषण करने के लिए किया गया है। ऐसे डेटा के विश्लेषण से, सुसंगतता कारक (α), डीकोहेरेंस कारक (p), बेरी चरण (ϕ_B), द्रव्यमान अवधि (m) जैसे विभिन्न टोपोलॉजिकल मापदंडों का अनुमान लगाया गया है।

भविष्य के इलेक्ट्रॉनिक उपकरणों में टोपोलॉजिकल मैटेरियल्स के कार्यान्वयन में बाधा डालने वाली चुनौतियों के संभावित समाधान सुझाने के बाद, कुछ बुनियादी सवालों के जवाब खोजने के लिए इस थीसिस कार्य में प्रस्तावित विकास पद्धति और जांच प्रोटोकॉल का उपयोग किया गया है। इनमें से कुछ प्रश्न हैं - चॉकोजेनाइड्स (टेल्यूरियम (Te) या सेलेनियम (Se)) की सांद्रता को ट्यून करके टोपोलॉजिकल मैटेरियल्स के टोपोलॉजिकल गुणों का प्रभाव क्या है, और टोपोलॉजिकल सामग्रियों के विकास के दौरान चुंबकीय और गैर-चुंबकीय तत्वों का समावेश क्या है? इन प्रश्नों का उत्तर देने के लिए, सबसे पहले, विभिन्न Te संरचना Bi_xTe_y बाइनरी मिश्र धातु पतली फिल्मों पर संरचनात्मक गुणों की जानकारी का उपयोग करके, नमूनों को तीन अलग-अलग Te-एकाग्रता चरणों में विभाजित किया जाता है, अर्थात् Te-deficient (TD), Te-stoichiometric (TS), और Te-excessive (TE)। इन तीन नमूनों पर मैग्नेटो-ट्रांसपोर्ट माप का उपयोग करके यह दिखाया गया है कि Te-एकाग्रता की ट्यूनिंग के माध्यम से, कोई टोपोलॉजिकल मैटेरियल्स के टोपोलॉजिकल गुणों में हेरफेर कर सकता है। इन प्रयोगात्मक निष्कर्षों को प्रथम सिद्धांत सैद्धांतिक गणना द्वारा आगे समर्थित किया गया है। दूसरी जिज्ञासा के लिए, Bi_2Te_3 फिल्मों को चुंबकीय तत्व के रूप में क्रोमियम (Cr) और गैर-चुंबकीय तत्व के रूप में सोना (Au) के साथ जोड़ा गया था। संशोधित एचएलएन और लू-शेन मॉडल का उपयोग करके बिना डोपड, Cr-डोपड और Au-डोपड Bi_2Te_3 फिल्मों पर मैग्नेटो-ट्रांसपोर्ट डेटा का प्रदर्शन और विश्लेषण करके विभिन्न टोपोलॉजिकल मापदंडों का अनुमान लगाया जाता है। अनुमानित मापदंडों के आधार पर, यह प्रदर्शित किया गया है कि बिना डोपड Bi_2Te_3 फिल्मों की तुलना में Cr-डोपड Bi_2Te_3 फिल्मों में टोपोलॉजिकल गुणों को काफी हद तक दबा दिया गया है, जबकि Au-डोपड Bi_2Te_3 फिल्मों में थोड़ा बढ़ाया गया है। यह अध्ययन टोपोलॉजिकल सिस्टम में टाइम रिवर्सल समरूपता के टूटने के कारण चुंबकीय अशुद्धियों की उपस्थिति में टोपोलॉजिकल गुणों के दमन की अच्छी तरह से स्थापित घटना से संबंधित है।

इन बुनियादी सवालों के जवाब देने के बाद, स्पटरिंग तकनीकों का उपयोग करके निर्मित टोपोलॉजिकल मैटेरियल्स और फेरोमैग्नेट्स के हेटरोस्ट्रक्चर में स्पिन गतिशीलता और स्पिन-चार्ज रूपांतरण दक्षता की जांच करने के लिए थीसिस कार्य को आगे बढ़ाया गया है। इन हेटरोस्ट्रक्चर नमूनों पर फेरोमैग्नेटिक रेज़ोनेंस (एफएमआर) माप किए जाते हैं और उनमें $\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ (CFB) की रोमैग्नेटिक फिल्मों की मोटाई में परिवर्तन के साथ α_{eff} और $\mu_0 M_{eff}$ की भिन्नता की जांच की जाती है। स्पिन डायनेमिक परिणाम बताते हैं कि प्रभावी स्पिन मिश्रण चालन को इन-प्लेन एफएमआर में डम्पिंग में दो-मैग्नेट स्केटरिंग (टीएमएस) योगदान का कम करके आंका गया है। आउट-ऑफ-प्लेन एफएमआर के विश्लेषण से, यह पाया गया है कि कमरे के तापमान पर प्रभावी स्पिन मिश्रण चालन क्रमशः टीएमएस योगदान के साथ और उसके बिना 13.13 nm^{-2} और 4 nm^{-2} है। इसके अतिरिक्त, प्रभावी स्पिन-मिश्रण चालन मूल्य धीरे-धीरे घटते तापमान के साथ बढ़ता है, जो Bi_2Te_3 की सतह सुसंगतता लंबाई के अनुसार है। ये निष्कर्ष टोपोलॉजिकल सामग्रियों के निकट रखे जाने पर फेरोमैग्नेट्स के चुंबकीयकरण गतिशीलता पर स्पिन-ध्रुवीकृत टोपोलॉजिकल सतह स्थितियों के प्रभाव को दर्शाते हैं। स्पटर-विकसित फेरोमैग्नेट-टोपोलॉजिकल इंसुलेटर हेटरोस्ट्रक्चर की उत्कृष्ट गुणवत्ता और इसका कुशल स्पिन-चार्ज रूपांतरण प्रदर्शन भविष्य के चुंबकीय मेमोरी उपकरणों में टोपोलॉजिकल सामग्रियों के महत्व को प्रदर्शित करता है। स्पिन डायनेमिक्स मापन $\text{PdTe}_2/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ बाइलेयर्स पर भी किया जाता है। $\text{PdTe}_2/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ बाइलेयर में प्रभावी स्पिन मिश्रण चालन और स्पिन धारा घनत्व क्रमशः $\sim 2 \times 10^{20} \text{ m}^{-2}$ और 2.5 MA m^{-2} पाए जाते हैं, जो HM/FM में देखी गई तुलना में काफी बड़े हैं। फेरोमैग्नेट बाइलेयर, और आण्विक किरण एपिटैक्सी तकनीक निर्मित फेरोमैग्नेट-टोपोलॉजिकल मैटेरियल्स हेटरोस्ट्रक्चर के लिए रिपोर्ट किए गए साहित्य के बराबर है।

इस प्रकार, यह थीसिस कार्य एक उद्योग-अनुकूल स्पटरिंग तकनीक द्वारा टोपोलॉजिकल मैटेरियल्स को संश्लेषित करने के लिए एक विकास पद्धति का प्रस्ताव करता है जिससे टोपोलॉजिकल मैटेरियल्स की किस्मों की उच्च उन्मुख और अच्छी गुणवत्ता वाली फिल्में बनाने में बहुत अच्छी सफलता मिली। इसके अलावा, यह कार्य साहित्य में उपलब्ध विभिन्न सैद्धांतिक मॉडलों का उपयोग करके टोपोलॉजिकल इंसुलेटर सिस्टम के मैग्नेटोट्रांसपोर्ट डेटा की विस्तृत व्याख्या का एकीकृत और व्यापक विवरण प्रदान करता है, जो भविष्य में इस क्षेत्र के मौलिक और तकनीकी विकास दोनों के लिए महत्वपूर्ण है। इसके अलावा, यह थीसिस चॉकोजेनाइड्स रचनाओं को ट्यून करके और टोपोलॉजिकल सिस्टम के विकास के दौरान चुंबकीय और गैर-चुंबकीय तत्वों को शामिल करके सामग्री की टोपोलॉजिकल विशेषताओं जैसे क्वांटम गुणों में हेरफेर करने का एक प्रभावी तरीका प्रदर्शित करती है। अंत में, यह थीसिस स्पटर किए गए विकसित फेरोमैग्नेट-टोपोलॉजिकल मैटेरियल्स बाइलेयर्स में कुशल स्पिन-चार्ज रूपांतरण प्रदर्शन को प्रदर्शित करती है जो भविष्य के स्पिट्रॉनिक्स उपकरणों में टोपोलॉजिकल सामग्रियों को लागू करने का मार्ग प्रशस्त करती है।

Table of Contents

Certificate.....	i
Acknowledgments.....	ii
Abstract	iv
संरvii	vii
List of Figures.....	xiii
List of Tables	xxxi
Chapter 1: Introduction.....	1
1.1 Fundamentals of Spintronics.....	2
1.1.1 Background.....	2
1.1.2 Spin current.....	3
1.1.3 Generation of spin current.....	5
1.1.4 Detection of spin current.....	12
1.1.5 Manipulation of spin current.....	14
1.2 Literature survey on SOT.....	18
1.3 Theoretical background on topological materials.....	19
1.3.1 Spontaneous symmetry breaking theory.....	19
1.3.2 Quantum Hall Effect.....	21
1.3.3 Topology in Geometric space.....	23
1.3.4 Berry Phase.....	24
1.3.5 Topological Insulator.....	27
1.4 Major challenges and motivation of the thesis.....	32
1.5 Objectives of the thesis work.....	33
1.5.1 Organization of the Thesis.....	34
Chapter 2: Experimental Details.....	40
2.1 Thin Film Deposition.....	41
2.1.1 DC magnetron sputtering.....	41
2.1.2 Ion Beam sputtering.....	44
2.2 Characterization Techniques.....	46
2.2.1 X-ray Diffraction.....	46
2.2.2 X-ray reflectivity.....	54
2.2.3 Raman Spectroscopy.....	57
2.2.4 Energy Dispersive X-ray spectroscopy (EDX).....	59
2.3 Measurement techniques.....	60

2.3.1 Magneto-transport measurement.....	60
2.3.2 Ferromagnetic Resonance measurement (FMR).....	63
Chapter 3: Optimization of the Growth parameters for fabricating Thin Films of Topological materials by Sputtering Route	66
3.1 Introduction.....	67
3.2 Materials and methods: TI growth recipe	68
3.3 Structural Characteristics of Bi_xTe_y TMs.....	72
3.3.1 X-ray diffraction analysis.....	72
3.3.2 Raman analysis	73
3.3.3 Debye Scherrer and Williamson Hall analyses	74
3.3.4 X-ray reflectivity analysis.....	74
3.3.5 Elemental composition analysis.....	76
3.4 Growth of other topological systems	76
3.5 Conclusions.....	77
Chapter 4: Investigations of the Non-trivial properties of Sputtered Topological Insulators using Magneto-transport measurements.....	79
4.1 Introduction.....	80
4.2 Experimental	81
4.3 Results and Discussion	82
4.3.1 Structure and composition analysis.....	82
4.3.2 Longitudinal resistivity and Hall effect.....	86
4.3.2 Envisaging topological insulators through electron transport.....	89
4.3.2 Experiments on Magneto-conductance curve of Bi_2Te_3 : Topological properties	96
4.4 Conclusions.....	101
Chapter 5: Effect of Te-composition on the Magneto-transport behaviour of the Sputtered Bi_xTe_y Topological Insulator films.....	102
5.1 Introduction.....	103
5.2 Experimental Methods	103
5.3 Magneto-transport Results and Analyses.....	105
5.3.1 Structural and composition data of 30 nm Bi_xTe_y films.	105
5.3.2 Electron-transport.....	105
5.4 Discussion and Conclusion	119
Chapter 6: Influence of incorporating the Magnetic and Non-magnetic dopants (Cr & Au) on the topological properties of the sputtered Bi_2Te_3	122
6.1 Introduction.....	123
6.2 Experimental Methods	123

6.3 Results and Discussion	124
6.3.1 Structural Properties.....	124
6.3.1 Transport Properties.....	128
6.4 Conclusions.....	132
Chapter 7: Spin Dynamics Studies in the Sputter grown $\text{Bi}_2\text{Te}_3/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ (BT/CFB) heterostructures by Ferromagnetic Resonance technique	133
7.1 Introduction.....	134
7.2 Experimental Methods	135
7.3 Results and Discussion	135
7.3.1 Structural Analysis.....	135
7.3.2 Spin Dynamics	136
7.4 Conclusions.....	147
Chapter 8: Investigation of structural, magneto-transport and spin dynamics properties of sputtered grown layered type-II topological semimetal PdTe_2 films and $\text{PdTe}_2/\text{Co}_{60}\text{Fe}_{20}\text{B}_{20}$ heterostructures	148
8.1 Introduction.....	149
8.2 Experimental Methods	151
8.3 Results and Discussion	152
8.3.1 Structural Properties.....	152
8.3.2 Magnetotranport Properties.....	157
8.3.2 Spin dynamics Properties.....	163
8.4 Conclusions.....	172
Chapter 9: Summary and Future Perspective.....	174
9.1 Summary.....	175
9.2 Future Perspective.....	178
References.....	179
Publication in International Journals	203
Published/Accepted.....	203
Under Review/Submitted.....	203
Other Publications (not included in the thesis).....	203
List of Presentations in National/International Conferences	206
Author's Biography	208

List of Figures

Figure no.	Figure caption	Page no.
1.1	Schematic illustrating the flow of (a) pure charge current (b) spin polarized current (c) pure spin current densities in conductors. These configuration of the different direction of current and spin current densities shown in Figure (a-c) under time reversal symmetry are depicted in (d-e), respectively.	4
1.2	Schematic of magnetization (\vec{M}) behavior under the influence of (a) an effective magnetic field (\vec{H}_{eff}) and (b) radio frequency (rf) magnetic field (h_{rf}) applied perpendicular along to the \vec{H}_{eff} . The precession of \vec{M} gradually damps toward \vec{H}_{eff} as shown in Figure (a) while damping torque is compensated by h_{rf} such that result in ferromagnetic resonance condition as illustrate in Figure (b).	6
1.3	Generation of transverse spin current density (J_s) originated from charge current density (J_e) due to spin Hall effect.	8
1.4	Schematic diagram showing deflection of spin polarized current ($\vec{\sigma}$) in relativistic magnetic field \vec{B} generated from polarized scattering centre or positive charged nucleus (represent by plus symbol).	8
1.5	Types of spin dependent scatterings: (a) Intrinsic mechanism arises due to finite Berry phase. Extrinsic mechanisms comprise of (b) Side jump and (c) Skew scattering.	9
1.6	Dispersion relation or band structure (E vs. k_x) of (a) spin degenerate band or parabolic band (b) Rashba band (c) Topological surface states of topological materials. Here, E_+ or E_- is energy value at positive and negative k , respectively.	11
1.7	Top view of Fermi surface of spin non-degenerate band of (a) Rashba band and (c) TSSs band. Under the influence of non-zero electric field or finite charge current J_C there is a shift in the Fermi surface, which results in non-zero spin accumulation or spin current in (b) Rashba band and (d) TSS band. (e-h) demonstrate the reverse process (spin to charge current) meaning that a non-	12

zero spin accumulation or spin current J_s^y results in finite charge current. The shaded region denoted the shifting of Fermi surface due to finite charge current in Figure (b and d) and spin current in Figure (f and h).

- 1.8 A schematic describing the transfer of spin current from FM to NM through spin pumping phenomena. The DC (H) and rf magnetic field (h_{rf}) are applied to achieve FMR condition. A spin current deflects into charge current due to inverse spin Hall effect and thereby generate open circuit voltage (V_{ISHE}). 13
- 1.9 Schematic of spin transfer torque. (a) An unpolarized electron beam is polarized after passing RL or fixed layer. The direction of the spin polarization of electrons after they have traversed the FL is same as that of M_1 . After passing through the thin spacer layer, the spin polarized electrons then enter the FL layer and flip their spin state. Consequently, the magnetization state of FL layer (M_2) gets altered by spin transfer torque. 15
- 1.10 Schematic of spin orbit torque (SOT) effect. A transverse spin current (of density J_s) is generated and injected into the FM layer when a charge current (of density J_c) flows through the NM layer. The torque on the local magnetic moment in the FM layer can be divided into two parts: field like torque (τ_{FL}), which drives the moment into precession around the effective magnetic field, and damping like torque (τ_{DL}), which acts to return the moment to its equilibrium position. 16
- 1.11 Schematics of single bit (a) STT-MRAM and (b) SOT-MRAM devices 17
- 1.12 A and B are atomic lattice sites with a chemical potential E and can host only one spinless fermion. When these two lattice sites are bring close to each other, the one level system splits into two non-degenerate levels (E_+ (antibonding) and E_- (bonding)). (b) N number of atomic sites having lattice constant a . (c) Introducing a finite lattice vibration (Δ) results into two possible atomic arrangements, namely I and II Vacuum states or ground level configurations. Variation of energy density (ϵ) as function of ka for 3 cases: (d) atomic arrangement without any lattice vibration 20

(corresponding to configuration (b)), (f) atomic arrangement with lattice vibration (corresponding to configuration (c)), and (e) Energy density (ϵ) as a function of order parameter (in present case lattice vibration (quantified by Δ)). The atomic arrangement (b) lies on the zero energy which is symmetric point, whereas by introducing small amount of Δ , the ground state energy level can spontaneously undergo transition into either of the two vacuum states (I or II) which do not possess any symmetry (*i.e.*, the symmetry-breaking). Hence, figure (e) depicts spontaneous symmetry breaking concept using ϵ vs. Δ relation.

- 1.13 Experimental results of (a) Quantum Hall effect (QHE) and (b) Fraction QHE in the form of R_{xx} or R_{xy} vs. Magnetic field (B) data. The *filling factor* (ν) associated with quantum Hall state (*i.e.*, $\sigma_{xx} = \nu e^2/h$) in Figure (a) is shown as top horizontal axis whereas in case of fraction QHE (Figure (b)), it is represented by the boxes and the numbers therein. 22
- 1.14 (a) A one-dimensional curved path with vectors \vec{v}_1 and \vec{v}_2 at positions 1 and 2, respectively. (b) Two-dimensional curved surface showing how a vector, initially at position 1, is parallel transport in cyclic route (via 1->2->3->4->1) to again reach at the same position, but with different angle. The difference of the angle between the initial and final vector orientation is represented by α which is also known as *anholonomy angle*. (c) Two-dimensional cyclic path on the surface of a three-dimension object, *i.e.*, sphere. Here α is equal to solid angle subtended by a cyclic two-dimensional curved path. 24
- 1.15 Atomic arrangements in the two lattice configurations (a) I Vacuum (b) II Vacuum. The bond strength of intracell is greater than that of intercell in I vacuum ($v > w$) while opposite ($w > v$) is true for II vacuum. 27
- 1.16 The trace of tip of vector \vec{h} when k changes from 0 to 2π for the two vacuum states: (a) I ($v > w$) and (b) II ($w > v$). 29
- 1.17 (a) The lattice configuration II Vacuum is sandwiched between two I Vacuum type lattice configurations. The interface of I-II or II-I is represented by dotted line. (b) Extreme case of Figure (a) 29

where for I Vacuum, the condition changed to $v > 0$; $w = 0$ while for II Vacuum, condition $w > 0$; $v = 0$ is imposed.

- 1.18 (a) Variation of $m = v - w$ with position x . (b) Typical band structure of a topological insulator. The states between the gap are topological surface states having linear or Dirac characteristics. (c) The confinement of electrons in presence of high magnetic field in 2D quantum Hall system. It illustrates the insulating nature of bulk states analogous to the topological bulk states. (d) Spin polarized states at the top and bottom edges of 2D quantum Hall system. (e) Schematic describing the helical spin polarized property of TSS on one of the surfaces of 3D TI (f) Schematic illustrating how backscattering in presence of non-magnetic impurity is forbidden in TSSs. 31
- 2.1 Actual image of *Excel instrument* Make sputtering system which is used to deposit topological materials. The different components associate to sputtering processes are accordingly labelled. (a) Inside view of the chamber: Sputtering guns and substrate holder inside the chamber (b) Inside view during plasma formation and sputtering. 43
- 2.2 The actual inner view of *NORDIKO* make the ion-beam sputtering deposition system (model: 3450). (b) Schematic of the same sputtering for better illustrating different components used for plasma extraction and sputtering process. The components related to the system are accordingly labelled in both figures. 45
- 2.3 (a) Actual image of *Malvern PANalytical* make x-ray diffractometer (Model-*Empyrean*). The different components related to diffractometer such as assembly of x-ray source, incident x-ray beam optics, goniometer, 5-axis sample state, Diffracted beam optics and x-ray detector assembly. (b) Schematic diagram of the interaction of x-ray with atomic planes which act as coherent scatters. 47
- 2.4 (a) Schematic of movement of x-ray source and detector in $\theta - 2\theta$ mode or Gonio mode (b) Schematic of Bragg-Brentano geometry in which the x-ray source and detector move over a goniometer 49

	circle. magnetization loops recorded along with the EA. Lines connecting the data points are guides to the eye.	
2.5	Schematic illustrating penetration of x-ray depth inside the film and substrate in case of (a) Gonio scan and (b) GIXRD scan. Angular displacement of source and detector in case of (c) Gonio and (d) GIXRD scans.	50
2.6	(a) Schematic of dislocation defects (b) Texture film in which crystallites containing different mosaic blocks (c) Angular spread in the normal vector \vec{H}_{hkl} (d) Geometry of goniometer and sample for the rocking curve (e) Spread in the width of diffracted peaks in I vs ω spectrum.	51
2.7	Schematic of Diffractometer geometry during phi scan measurements in which first source and detector kept constant at Bragg angle of probed plane and then (a) sample stage is rotated to angle χ and after that (b) rotated full $360^\circ \phi$ angle.	52
2.8	(a) Relation between the diffraction vector and the angle of incidence and detection. (b) Schematic of diffraction space coordinates (Q_x, Q_z) representation or RSM plot for case (b) when two different phases (or structurally different film and substrate) have dissimilar d_{hkl} values, and (c) when the mosaic domains of different phases have different tilts. In (b) and (c), the two sets of circular rings correspond to the two different phases. Different colored rings within one such sets represent the spread in d_{hkl} (in (b)) and mosaicity (in (c)) in that particular phase. For clarity, the spread is assumed only in one direction, due to which the contours are circular and the two sets are either one over each other (in (b)) and side by side (in (c)). In case of ideal phases, the two sets would overlap to show just one small spot.	53
2.9	Schematic of XRR scattering geometry which results in specular and diffuse reflection from the rough, thin film surface	54
2.10	(a) Sketch showing the angles and indices of refraction along with the wave vector directions when x-rays are incident from the air on the surface of a sample having a refractive index less than 1. (b) Specular XRR profile of 15 nm thick CoFeB film deposited on	56

	Si (100) substrate indicating the information provided by x-ray reflectivity profile.	
2.11	(a) Schematic of the Raman measurements. (b) Raman spectra measured for Bi ₂ Te ₃ (40 nm) thin film. The spectra is fitted to show two E _g ² , and A _{1g} ² Raman modes.	58
2.12	Schematic diagram showing the working principle of EDX	59
2.13	(a) Actual image of <i>Quantum Design</i> make PPMS (Model-6000 <i>Evercool II</i>) (b) Inner view of PPMS probe (c) Actual image of <i>Cryogenic Ltd.</i> make CFMS (d) Schematic diagram of VTI cooling circuit.	60
2.14	(a) Four probe resistivity measurement geometry (b) Four terminal Hall measurement geometry	61
2.15	Schematic of magnetoresistance measurement (MR) geometry for (a) Out of plane magnetic field configuration and In-plane field configuration in (b) Transverse (c) Longitudinal in-plane geometry.	63
2.16	Schematic of ferromagnetic resonance setup (a) in-plane field configuration (IPFMR) and (b) Out-of-plane field configuration	64
3.1	Cross-sectional schematic and interior view of the DC/RF Excel instrument make DC/RF magnetron sputtering which is used for the growth of TMs using the co-sputtering technique.	69
3.2	(a) and (b) X ray reflectivity (intensity vs. 2θ) spectra recorded on Bi and Te samples deposited at different powers to the Bi and Te sputtering targets, respectively keeping deposition time fixed at 5 mins. The symbols represent the experimental data and solid lines are simulated fits. (c) The solid symbols represent growth rate at different powers. The data is fitted to straight line to estimate the slope (i.e., growth rate per Watt) for both Bi (square) and Te (sphere) films (d) Nominal (calculated using Equation 3.1 and 3.2) and experimentally determined (observed using EDAX) Te-to-Bi atomic concentration ratio (y/x) in Bi _x Te _y films by co- sputtering as a function of power applied to Te sputtering target (Power to the Bi-target was kept a constant value of 5W). cross-sectional	71

- schematic and interior view of the DC/RF *Excel instrument* make DC/RF magnetron sputtering which is used for the growth of TMs using the co-sputtering technique.
- 3.3 (a) Atomic schematic corresponding to Te deficient to excessive concentration phases in Bi_xTe_y system. (b) XRD spectra recorded in Bragg-Brentano mode (*i.e.*, $\theta - 2\theta$ coupled scan) of all the Bi_xTe_y thin films, where x and y are the atomic concentrations. The experimental data is represented by symbols and lines represent the fit. Different symbols used are \blacktriangle for Bi_4Te_3 , \blacklozenge for $\text{Bi}_{50}\text{Te}_{50}$, \spadesuit for Bi_4Te_5 and \bullet for Te. The vertical dotted lines correspond to (0 0 1) peaks of the Bi_2Te_3 phase. (c) Raman spectra of the Bi_xTe_y films. The two vertical dotted lines corresponds to the E_g^2 and A_{1g}^1 Raman active modes of Bi_2Te_3 . The peaks identified by symbol \bullet correspond to the Raman shift of Te. 72
- 3.4 Variation of (a) lattice parameter 'c', (b) crystallite size and strain with relative Te atomic percentage in Bi_xTe_y films. (c) XRR spectra (intensity vs. 2θ) of the different Bi_xTe_y thin films. The symbols represent the experimental data and red lines show the simulated fittings. Variation of (d) roughness σ and (e) density ρ with the Te concentration in Bi_xTe_y films. The model used for simulation is shown in the inset of (d). 75
- 3.5 EDAX elemental mapping of samples (i) $\text{Bi}_{49}\text{Te}_{51}$, (ii) $\text{Bi}_{34}\text{Te}_{66}$ and (iii) $\text{Bi}_{27}\text{Te}_{63}$. Here, (a) shows mixed homogenous composition of Bi (red dots) and Te (green dots). (b) and (c) show Bi and Te maps, separately. 76
- 3.6 X-ray diffraction spectrum of (a) NiTe_2 (b) PtTe_2 (c) PdTe_2 (d) PdBi_2 (e) $\text{Bi}_x\text{Sb}_{1-x}$ (f) Cr_2Te_3 77
- 4.1 (a) $\theta - 2\theta$ XRD pattern of Bi_2Te_3 film (b) Rocking curve scan along the (0 0 6) Bragg planes. Here symbols are the experimental data and lines are fits. (c) The ϕ -scan along the (0 1 5) Bragg planes of Bi_2Te_3 film and (0 1 -1 2) Bragg planes of Al_2O_3 substrate. Inset: Schematic of layered Bi_2Te_3 on top of Al_2O_3 . (d) Reciprocal space mapping of the film is performed along the (0 0 1 5) planes of Bi_2Te_3 and (0 0 6) planes of Al_2O_3 . 83

- 4.2 (a) Raman spectra (Intensity vs. wavenumber (ν)). The two Raman active modes of Bi_2Te_3 , namely E^2_g and A^2_{1g} are labelled above the corresponding Raman peaks. (b) XRR spectra of the sample. The model used for the simulation and the values of the thickness, density, and roughness so-obtained from the simulation are also presented within the main panel itself. The numbers in the parenthesis represent the error. Figure 4.1(a & b) symbols are the raw experimental data, and the fitting is represented by solid lines. 84
- 4.3 Elemental mapping of (a) Bismuth and (b) Tellurium elements on $\sim 50\mu\text{m}^2$ area of the sample (c) EDAX spectrum (counts/sec vs keV) for Bi and Te atoms showing the wt.% of Bi_2Te_3 film (To remove the contribution of substrate (Al_2O_3), EDS spectrum for Al and O atoms is excluded while accumulating the spectrum). The table in the inset of (c) shows composition in weight and atomic percentage of Bi and Te atoms in Bi_2Te_3 film (σ denotes the standard deviation in estimation of weight percentage). 85
- 4.4 (a) Sheet resistivity as a function of T (4-300K) of the Bi_2Te_3 film. Inset shows the data at low T (4-15K) (b) Hall resistivity with the magnetic field recorded at different T . Sheet carrier concentration (solid circle symbols) and mobility (solid square symbols) at different T are shown in the inset. The T dependence of ρ of the Bi_2Te_3 film in (c) low T (4-15 K) and (d) High T (25-300K) regimes. The low T data (Figure (c)) is fitted using Equation 4.1 ($\rho = \rho_o + \rho_{e-p}(T) + \rho_{EEI}(T)$) and high T data (Figure (d)) is fitted using Equation 4.2 ($\rho = \rho_o + \rho_{linear}(T) + \rho_{non-linear}(T)$). The experimentally (Expt.) observed data is shown by open circle symbols while the fitted data are shown by solid lines. The T dependence of individual contribution related to different scattering terms (i.e., ρ_{e-p} , ρ_{EEI} , ρ_{linear} and $\rho_{non-linear}$) which are estimated from the fitting are shown by solid lines. 88
- 4.5 (a) The out of plane magnetic field (B_{\perp}) dependence of the sheet conductivity correction (in units of $\frac{e^2}{h}$) at different constant Ts (3-20K). (b) The variation of L_{ϕ} corresponding to the surface states (filled circle) and bulk states (filled square) with the T. The 96

- symbols represent the experimental data, and lines are the fit using to Equations (4.5) and power law $L_{\phi_s} \propto T^{-\frac{p_s}{2}}$, respectively.
- 4.6 (a) Semi-log plot of sheet conductivity (in units of $\frac{e^2}{\pi h}$) with the logarithmic T at different out-of-plane applied magnetic field strengths. The solid lines are the fit to experimental data using $\kappa = \frac{\pi h}{e^2} \frac{\delta\sigma}{\delta \ln T}$. (b) The out-of-plane magnetic field dependence of κ parameter. The characteristics T of the EEI effect (T_{EEI}) is found to be 10K. 99
- 4.7 (a) Variation of sheet conductivity (in units of $\frac{e^2}{\pi h}$) with the applied in-plane magnetic field at different constant T (3-15K). (b) The variation of surface penetration depth λ with T and the T dependence of the calculated value of ratio (t_2/t_1) is shown in the inset of Fig. (b). 100
- 5.1 (a) XRD spectra recorded in Bragg-Brentano mode (*i.e.*, $\theta - 2\theta$ coupled scan) for TD, TS and TE samples. The experimental data are represented by symbols. While \blacklozenge symbols denotes peaks due to $\text{Bi}_{50}\text{Te}_{50}$ phase, \bullet represents the peaks due to Te. The vertical dotted lines correspond to (0 0 l) peaks of the Bi_2Te_3 phase. (b) XRR spectra (intensity vs. 2θ) of the TD, TS and TE samples. The symbols represent the experimental data and red lines show the simulated fittings. (c) Raman spectra of the TD, TS and TE samples. The two vertical dotted lines correspond to the E_g^2 and A_{1g}^1 Raman active modes of Bi_2Te_3 . The peaks identified by the symbol \bullet correspond to the Raman shift due to Te. (d) The atomic composition (in at%) for TD, TS and TE samples extracted using EDX. 104
- 5.2 Electron transport behavior in TD, TS and TE samples. (a) Resistivity vs. temperature data. (b) Out-of-plane magnetoresistance data and (c) Hall resistivity as a function of magnetic field ($\mu_0 H$) at 3K temperature. Measurement geometries are shown in the insets. 106
- 5.3 Logarithmic resistivity as a function of $1/T$ for (a) TD, (b) TS and (c) TE samples. Lines represent the fit using the Arrhenius 107

equation, $\rho_{xx} \sim \exp(E_a/k_B T)$ in the high temperature regime (200-300K). The temperature dependence of resistivity in the low-temperature range (3-15K) at 0T (filled circle symbols) and 9T (open square symbols) for (d) TD, (e) TS and (c) TE samples. The data at 0T and 9T correspond to WL/WAL+EEI and EEI, respectively

- 5.4 Hall resistivity vs magnetic field at different temperatures for the samples (a) TD, (b) TS and (c) TE. Inset in (b) shows the schematic of the measurement geometry. Symbols represent the experimental data points and the solid lines are the linear fits. Sheet carrier concentration and mobility with temperatures for samples (d) TD, (e) TS and (f) TE. Note the change in the order of magnitude of mobility in the three films, as is evident from the y-axis title. 114
- 5.5 Variation of sheet conductivity ($\Delta\sigma$), in units of $\frac{e^2}{h}$, with the applied out-of- plane magnetic field (B_{\perp}) at different constant temperatures (3-10K) for the samples (a) TD, (b) TS and (c) TE. Solid lines are fit to the experimental data using Equation 5.6. (d-e) and (g-i) T dependence of the pre-factor (α) and phase coherence length (l_{ϕ}), respectively, calculated using fit to data in (a-c) by Equation (5.3). (g-i) also show the variation of the calculated l_{ϕ} (open circles), l_e (filled squares) and l_{so} (open triangles) with temperature. The solid lines show fitting of the l_{ϕ} data by power law mentioned explicitly in the insets. 116
- 5.6 Plots of logarithmic dependence of sheet conductivity (in units of $\frac{e^2}{\pi h}$) vs. temperature at different perpendicular magnetic field strengths for the samples; (a) TD, (b) TS and (c) TE. The symbols represent the experiment data and solid lines show the fits using $\kappa = \frac{\pi h}{e^2} \frac{\delta\sigma}{\delta \ln T}$. The dependence of κ parameter on the magnetic field for (d) TD, (e) TS and (f) TE samples 118
- 5.7 Sheet conductivity $\Delta\sigma$ (in units of $\frac{e^2}{h}$) with in-plane magnetic field (B_{\parallel}) recorded at 3K for the TD, TS and TE samples. Lines are fit using Equation (5.8). (b) & (c) The variation of pre-factor (α_{\parallel}) and 120

- surface penetration depth (λ), calculated from fitting of $\Delta\sigma$ vs μ_0H for the range of temperatures. (d) Calculated values of t_2/t_1 ratio as a function of T for the TD, TS and TE samples.
- 5.8 Layer slabs for three different compositions of Bi_xTe_y used in calculations and the simulated Brillouin Zones (BZ's) for bulk and surface states. Calculated electronic band structures of (b) BiTe (TD), (c) Bi_2Te_3 (TS), and (d) BiTe_3 (TE). Fat bands of surface states are plotted with different colors. Insets are the zoomed version of the Dirac-cone and surface states near the Fermi level. The size of the filled circles is proportional to the weight of the surface states. 120
- 6.1 (a) XRD spectra (Intensity vs. 2θ) recorded in the $\theta - 2\theta$ coupled scan, also known as Gonio mode. The dashed-dotted lines at peak positions are used to label Bragg planes of Bi_2Te_3 identified using JCPDS #894302. The unlabelled peaks belong to Si (100) substrate. (b) The peak along (006) Bragg plane is fitted using the pseudo-Voigt function, which assumes the peak profile as a linear combination of Gaussian and Lorentzian peak curves. The center position and FWHM of the peak are extracted using the fitted data. The calculated lattice parameter ('c-axis') and crystallite size of undoped, Au doped and Cr doped samples are mentioned on the inset of the graphs. The symbol represents the raw experimental data, and the solid line represents the fitting. 125
- 6.2 (a) Raman spectra (Intensity vs. wavenumber (ν)) of undoped, Au doped, and Cr doped samples. The three Raman active modes of Bi_2Te_3 , namely E_g^2 , A_{1g}^2 , and A_{1u} labeled above the corresponding Raman peaks. The pseudo-Voigt function fits the raw experimental data (shown by symbol) to extract FWHM and peak position information. (b) XRR spectra (Intensity vs. 2θ) of the undoped, Au-doped, and Cr-doped samples. Here, the symbol represents the raw experimental data while the solid line represents the fitted data. (c) The model used for simulation of XRR data using X'pert reflectivity software and thickness, density, and roughness is estimated as shown in Table 6.1 126

- 6.3 (a) EDAX elemental mapping of samples (i) undoped, (ii) Au doped and (iii) Cr doped. Here, (a) shows the mixed homogenous composition of Bi (red dots), Te (green dots), and doping element (blue dots). (b), (c) and (d) show Bi, Te and Au or Cr maps separately 127
- 6.4 Variation of sheet resistivity with temperature (2-300K) for samples (a) undoped, (b) Au doped and (c) Cr doped, respectively. 128
- 6.5 (a) Variation of sheet conductivity ($\Delta\sigma$), in units of $\frac{e^2}{h}$ With the applied out-of-plane magnetic field (B_{\perp}) at different constant temperatures (2-20K) for the (a) undoped, (b) Au doped and (c) Cr doped samples. Solid lines are fit to the experimental data using Equation (6.1). (d-e) Temperature dependence of the pre-factor (α) and (g-i) Temperature dependence of the phase coherence length (l), respectively, calculated using fit to data in (a-c). (g-i) also show the variation of the calculated l_s (open circle symbol) and l_b (open square symbol) with temperature. The solid lines show the fitting of the l_s data by a power law. 130
- 6.6 (a) Variation of sheet conductivity $\Delta\sigma$ (in units of $\frac{e^2}{\pi h}$) with the logarithmic temperature at different out-of-plane applied magnetic field strengths. The solid lines are the fit to experimental data using $\kappa = \frac{\pi h}{e^2} \frac{\delta\sigma}{\delta \ln T}$. (b) The out-of-plane magnetic field dependence of κ parameter. The characteristics temperature of the EEI effect (T_{EEI}) is found to be 10 K for undoped and Au doped and 15 K for Cr doped samples. 131
- 7.1 $\theta - 2\theta$ XRD pattern of $Al_2O_3//BT(10)/CFB(18)/Al(5)$ sample. Diffraction peaks only due to the $(00l)$ family of Bragg planes of BT are observed. The peak due to substrate is marked as $Al_2O_3(006)$. The peaks labelled by * symbol arise due to contamination in X-ray source and imperfection in crystallographic arrangement of Al_2O_3 (b) The XRR pattern (intensity vs. 2θ) of $BT(10)/CFB(t_{CFB})$ series and reference CFB(10) sample. The open symbols represent the experimental data points, while solid lines are simulated fits. 136

- 7.2 (a) Variation in $\mu_0 H_r$ as a function of microwave frequency for samples of series (a) CFB (t_{CFB}) (b) BT (10)/CFB (t_{CFB}), respectively. The points are the data extracted from the fitting of IPFMR spectrum, and the lines are fits corresponding to the in-plane Kittel Equation (7.2) (b) The variation in $\mu_0 M_{eff}$ for all the samples in series CFB (t_{CFB}) and BT (10)/CFB (t_{CFB}) as a function of t_{CFB} . 137
- 7.3 $\mu_0 \Delta H$ vs. frequency for series (a) CFB (t_{CFB}) and (b) BT (10)/CFB (t_{CFB}). The symbols represent the experimentally obtained values, which were fitted with Equation (7.3) to obtain the effective damping constant (α_{eff}) and the inhomogeneous linewidth ($\mu_0 \Delta H_0$). (c) α_{eff} vs. t_{CFB} for BT (10)/CFB (t_{CFB}) (open circle symbols) and CFB (t_{CFB}) (open square symbols). 138
- 7.4 (a) The Effective damping (α_{eff}) vs. inverse of CFB thickness for BT(10)/CFB(10) heterostructure. (b) The dependency of spin current density (J_s) of series BT (10)/CFB (t_{CFB}) with the t_{CFB} layer for different frequency ranges (4-12 GHz) represented by a symbol, and solid lines connecting the data points guide the eye. 139
- 7.5 (a) Out-of-plane FMR signal for BT(10)/CFB(10) heterostructure at 5K and 300K. (b) Linewidth vs. frequency for bare CFB(10) and BT(10)/CFB(10) heterostructure at 5K and 300K. (c) Frequency vs. resonance field for bare CFB(10) and BT(10)/CFB(10) heterostructure at 5K and 300K. Inset shows the temperature dependence of the effective magnetization. (d) Damping parameter (α) as a function of temperature for bare CFB(10) and BT(10)/CFB(10) heterostructure. 142
- 7.6 Temperature dependence curve of α_{eff} (open square symbol), α_{eddy} (open rhombohedral symbol), α_{rad} (open hexagonal symbol), and α (open circle symbol), for sample (a) CFB (10) and (b) BT (10)/ CFB (10). 143
- 7.7 (a) The real part of effective spin mixing conductance as a function of temperature for BT(10)/CFB(10) heterostructure. (b) The magnetic field ($\mu_0 H$) dependent magnetoconductance (MC) correction ($\Delta\sigma$) (in units of $\frac{e^2}{h}$) response of BT sample at different 146

constant temperatures (3-10K). The symbols represent the experimental data, and lines are the fits using the modified HLN model. (c) Temperature dependence of L_ϕ corresponding to the surface states (filled circles) and bulk states (filled squares). The solid lines are fits using the power law $L_{\phi_s} \propto T^{-n}$. (d) $\Delta\sigma$ (in units of $\frac{e^2}{h}$) vs. magnetic field ($\mu_0 H$) of BT(10)/CFB(10) heterostructure at 2K, 4K, 6K and 10K

- 8.1 (a) XRD spectra (Intensity vs 2θ) recorded in the Gonio mode (i.e., $\theta - 2\theta$ mode) for PT (10), PT (20) and PT (50) films. The Bragg planes are labelled for the corresponding diffraction peaks. For clarity and clear comparison, a break is inserted in the x-axis from 36-45 degrees to hide the highly intense (006) Bragg peak originating from the single crystalline Al_2O_3 substrate. (b) Omega (ω) scan or rocking curve scan (Intensity vs ω) on PT (10), PT (20) and PT (50) films. (c) Raman spectra (Intensity vs. Raman shift) for the PT (10), PT (20) and PT (50) films with the corresponding Raman modes (E_g and A_{1g}) mentioned adjacent to the Raman peaks. (d) X-ray Reflectivity (XRR) spectra (Intensity vs 2θ) for PT (10), PT (20) and PT (50) films. The open square symbols, open circle symbols and open triangle symbols are used to illustrate the experimental data of PT (10), PT (20) and PT (50) films, respectively while solid lines show the fitting of data with mentioned functions and models. 153
- 8.2 Unit cell along and the side views of 4×4 superlattice structure of (a) PtTe_2 and (d) NiTe_2 where Pt and Ni atoms occupy corner positions while Te sits at the body diagonal of the unit cell. XRD spectra in the Gonio mode ($\theta - 2\theta$) for (b) PtTe_2 and (e) NiTe_2 films. The Bragg planes are labelled corresponding to the observed diffraction peaks. Raman spectra (Intensity vs. Raman shift) recorded on the (c) PtTe_2 , and (f) NiTe_2 films. The corresponding Raman modes are mentioned above the respective Raman peaks. The filled circles symbols represent the experimental data whereas the solid lines are the fit using the respective peak functions. 155

(pseudo-Voigt peak function for XRD, and Lorentzian function for Raman).

- 8.3 (a) Temperature-dependent resistivity (ρ vs T) curves from 2-300 K of (a) PT (10) (b) PT (20) and PT (50) samples, where open data symbols represent the experimental data while the solid lines represent the fitted ρ vs T curves using Equation (8.1) from ~15-300K. The inset of the figures shows ρ vs T curves from 2-10 K. The T_{onset} and T_C temperatures correspond to superconductivity temperatures mentioned in the inset of the figures (The solid lines connecting the data points in the inset s are guide to the eye). 159
- 8.4 Variation of magnetoresistance ($MR(\%) = \frac{R(B) - R(B=0)}{R(B=0)} \times 100$) with magnetic field ($\mu_0 H$ in units of T) between -3T to 3T at 3K, 5K, 7K and 10K for (a) PT (10) (b) PT (20) and (c) PT (50) samples, respectively. In figures (a-b), while the open symbols represent the experimental data, the solid lines show the fitting of high field MR data using $MR(\%) = AB^n$. The insets in figures (a-c) show the $MR(\%)$ vs $\mu_0 H$ curve at 2K for samples PT (10), PT (20) and PT (50), respectively. (d) and (e) show the variation in the correction in the sheet conductivity ($\Delta G = \frac{1}{\rho_s(\mu_0 H)} - \frac{1}{\rho_s(0)}$ in units of $e^2/\pi h$) with the magnetic field ($\mu_0 H$) at 3K, 5K, 7K and 10K for samples PT (10) and PT (20), respectively. Here the experimental data are represented by open symbols while the solid lines depict the fitting of $\Delta\sigma$ vs $\mu_0 H$ curve using the modified HLN model given by Equation (8.2). (f) Schematic of measurement geometry for MR. Here, V, I_x and H_z imply the voltmeter, source current flowing along the samples surface and transverse magnetic field applied along the z axis. 160
- 8.5 (a) Temperature-dependent variation of coherency prefactor (α) for samples PT (10) and PT (50). The open square and open circle symbols represent the estimated values of α for PT (10) and PT (50) films, respectively at different temperatures between 3-10 K. (b) Temperature-dependent variation of phase coherence length (L_ϕ in units of nm) for samples PT (10) and PT (50). The open square and open circle symbols represent the calculated values of 162

L_ϕ for PT (10) and PT (50) films, respectively at different temperatures between 3-10 K while solid lines describe the fits using the power law equation ($L_\phi \propto T^{-\frac{p}{2}}$) where p is the dephasing parameter. For the PT (10) sample, p is 1.46 while for PT (20), p is 1.66, respectively.

- 8.6 (a) Magnetic field dependent variation of FMR signal collected at 7 GHz for (a) series-I (PT (0)/CFB (t_{CFB})), (b) series-II (PT (10)/CFB (t_{CFB})), (c) series-III (PT (20)/CFB (t_{CFB})) and (d) series-IV (PT (50)/CFB (t_{CFB})) samples, where $t_{CFB} = 10, 15$ and 20 nm. Here, the open square, open circle and open triangle symbols denote experimentally obtained data of PT (t_{PT})/CFB (10), PT (t)/CFB (20) and PT(t_{PT})/CFB (50) films ($t = 0, 10, 20$ and 50 nm), respectively. The solid lines represent the fitted derivative FMR signal *vs* $\mu_0 H$ curve using Equation (8.3). 163
- 8.7 Variation of resonance field ($\mu_0 H_r$ in units of mT) as a function of microwave frequency (f in units of GHz) for (a) series-I (PT (0)/CFB (t_{CFB})), (b) series-II (PT (10)/CFB (t_{CFB})), (c) series-III (PT (20)/CFB (t_{CFB})) and (d) series-IV (PT (50)/CFB (t_{CFB})) samples, respectively. Here the open square, open circle and open triangle symbols are the data extracted from the fitting of the FMR spectrum of PT (t_{PT})/CFB (10), PT (t_{PT})/CFB (20) and PT (t_{PT})/CFB (50) films (where $t_{PT} = 0, 10, 20$ and 50 nm), respectively, and lines are the fits corresponding to the Kittel Equation (8.4). 165
- 8.8 (a) The variation in $\mu_0 M_{eff}$ for all the series I-IV as a function of t_{CFB} . Here the open square, open circle, open triangle and open diamond symbols are the values of $\mu_0 M_{eff}$ estimated from the fitting of the FMR spectrum of the series-I (PT (0)/CFB (t_{CFB})), series-II (PT (10)/CFB (t_{CFB})), series-III (PT (20)/CFB (t_{CFB})) and series-IV (PT (50)/CFB (t_{CFB})) samples, ($t_{CFB} = 10, 15$ and 20 nm) respectively, using Equation (8.4). The solid lines are provided as a visual aid to demonstrate the overall variation. (b) $1/t_{CFB}$ dependence of $\mu_0 M_{eff}$ for all the series I-IV, here solid line depicts the fitting of $\mu_0 M_{eff}$ *vs* $1/t_{CFB}$ curve using Equation (8.5) 166

(c) Histogram plot of the thickness-dependent variation of anisotropic field (H_k) for series I-IV samples. Here, bars correspond to the values of $\mu_0 H_k$ as calculated using the fitting of the FMR spectrum for all series I-IV samples using Equation (8.4).

8.9 Variation of linewidth ($\mu_0 \Delta H$ in units of mT) with frequency (f in units of GHz) for (a) series-I (PT (0)/ CFB (t_{CFB})), (b) series-II (PT (10)/ CFB (t_{CFB})), (c) series-III (PT (20)/ CFB (t_{CFB})) and (d) series-IV (PT (50)/ CFB (t_{CFB})) samples, respectively. Here the open square, open circle and open triangle symbols are the values of $\mu_0 \Delta H$ estimated from the fitting of the FMR spectrum of PT (t_{PT})/CFB (10), PT (t_{PT})/CFB (20) and PT (t_{PT})/CFB (50) samples ($t_{PT} = 0, 10, 20$ and 50 nm), respectively. The solid lines are the fits corresponding to the linewidth given by Equation (8.6). 168

8.10 (a) CFB layer thickness dependent (t_{CFB}) variation of Gilbert damping constant ($\alpha \times 10^{-3}$) of the different samples of series I-IV. Here, the open square, open circle, open triangle and open hexagonal symbols represent the values of $\alpha \times 10^{-3}$ estimated from the fitting of $\mu_0 \Delta H$ vs f using linewidth Equation (8.6) for series-I (PT (0)/ CFB (t_{CFB})), series-II (PT (10)/ CFB (t_{CFB})), series-III (PT (20)/ CFB (t_{CFB})) and series-IV (PT (50)/ CFB (t_{CFB})) samples, (where $t_{CFB} = 10, 15$ and 20 nm) respectively. The solid lines are provided as a visual aid to demonstrate the overall dependency. (b) The variation of $\Delta \alpha \times 10^{-3}$ with respect to $1/t_{CFB}$. Here $\Delta \alpha$ is the finite-size contribution to damping, and is obtained from subtracting the respective α value of the samples of series-I (PT (0)/CFB (t_{CFB})) series from the corresponding values of α obtained for samples of the series PT (t_{PT})/CFB (t_{CFB}). The open circle, open triangle and open diamond symbols correspond to $\Delta \alpha \times 10^{-3}$ values for series-II ($t = 10$ nm), series-II ($t = 20$ nm) and series-III ($t = 50$ nm), respectively and $t_{CFB} = 10, 15$ and 20 nm. Additionally, the solid lines are the fit of the observed $\Delta \alpha \times 10^{-3}$ vs $1/t_{CFB}$ variation using Equation (8.7). (c) Histogram plot showing the FM thickness dependent (t_{CFB}) variation in the inhomogeneous broadening of the all the sample of series I-IV. 169

8.11 The dependency of spin current density (J_s) estimated using Equation (8.8) of the PT (10)/ CFB (t_{CFB}) series (open circle symbols), PT (20)/ CFB (t_{CFB}) series (open triangle symbols) and (c) PT (50)/ CFB (t_{CFB}) series (open diamond symbols) samples on t_{CFB} at 4 GHz. The solid lines connecting the data points are guide to the eye. 170

List of Tables

Table no.	Table caption	Page no.
1.1	Parameters associated with SOT-based switching in NM/FM heterostructure.	18
2.1	Optimised sputtering parameters for the growth of CFB and Al layers used in the different heterostructures.	46
3.1	The calculated (nominal) and observed (experimental) atomic composition as determined from XRR and EDAX measurements, respectively at different power applied to Te target (P_{Te}) with fixed power to Bi target ($P_{Bi}=5W$).	70
3.2	The XRR-simulated parameters such as thickness t , interface/surface roughness σ , and density D of the layers in Bi_xTe_y . The numbers in the parenthesis represent the error in the simulation.	74
4.1	The fitted values of the different parameters appearing in the T dependence of the resistivity of Bi_2Te_3 films.	70
4.2	Comparison of the MC response (Dominant or Suppressed) of conventional electrons and Dirac Fermions vis-à-vis the different types of disorder induced scattering such as scalar, magnetic impurity, and spin-orbit type.	74
4.3	The different band parameters describing the MC behaviour in Bi_2Te_3 film, as calculated using Equations (4.7a-c), (4.8), and (4.9).	98
5.1	Calculated values of sheet conductance (σ_s), sheet carrier concentration (n_s), mobility (μ), mean free path (l_e), dimensionless conductance (g), diffusion constant (β) and thermal length (l_T) for the sample TD, TS and TE at 3K temperature. For comparison, the reported data on MBE grown TIs and sputter-grown Bi_2Se_3 samples are also added. Symbol ‘*’ indicates that the values are calculated from the parameters provided in the given references.	111

6.1	The XRR-simulated parameters such as thickness t , interface/surface roughness σ , and density D of the layers in Bi_xTe_y . The numbers in the parenthesis represent the error in the simulation.	127
7.1	Simulated values of thickness, interface width, and density of each layer. The numbers in the parentheses represent the error in simulations.	136
7.2	The different parameters extracted from the IPFMR measurements of TIs-FM heterostructures (this work) and their comparison with relevant literature values. Numbers in the bracket represents the errors in the values.	141
8.1	Details of the specific sample stacks used in 4 different heterostructures series.	151
8.2	Values of various parameters extracted from the analysis of ρ vs T data for the PT films having different thickness. The values in the bracket corresponded to the errors in the fitting.	158
8.3	The extracted values of $\mu_0 M_s$ and K_s for the various samples of series I-IV. The values in the bracket corresponded to the errors in the fitting.	167
8.4.	Comparison of the different parameters viz., surface anisotropy, effective spin-mixing conductance and spin current density) as extracted from the FMR measurements of the PT/FM heterostructures (this work) with the relevant values cited in the literature references (last column). Numbers in the bracket represent the errors in the values. The thickness of TMDs (MoS_2 and WS_2) is specified in MLs which stand for number of monolayers.	171